

## Glass Formation in Multicomponent Phosphate Systems Containing $\text{TiO}_2$

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Glass formation was investigated in the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$ ,  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$ ,  $\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$ ,  $\text{Na}_2\text{O}-\text{BaO}-\text{P}_2\text{O}_5$ , and  $\text{Na}_2\text{O}-\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$  systems. In the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  system, glass containing up to 45.5 mol%  $\text{TiO}_2$  was found to be formed at the composition of  $\text{Na}_2\text{O}/\text{P}_2\text{O}_5=1$ , whereas in the  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  system the glass formation was observed in a narrow region along the composition of  $\text{K}_2\text{O}/\text{P}_2\text{O}_5=1$  in a three-component diagram. In the  $\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$  system, only the glass containing about 10 mol%  $\text{TiO}_2$  as a maximum was obtained. In the  $\text{Na}_2\text{O}-\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$  system, the region of glass formation extended with an increase of  $\text{Na}_2\text{O}$  up to 30 mol%. These results suggested that a certain amount of  $\text{Na}_2\text{O}$  is necessary for the formation of glass containing a large amount of  $\text{TiO}_2$  in the presence of  $\text{BaO}$ . The three-component diagrams of the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  and  $\text{Na}_2\text{O}-\text{BaO}-\text{P}_2\text{O}_5$  systems indicated that the most favorable amount of  $\text{Na}_2\text{O}$  for the glass formation in this four-component system is 27 to 28 mol%.

The region for glass formation in a three-component diagram is determined by the role of cations in the glass structure. The conditions and the region for glass formation in ternary systems of borate, silicate, germanate, and tellurite have been studied in detail by Imaoka *et al.*<sup>1,2)</sup> However, there have been few investigations of glass formation in multicomponent phosphate systems of three or more components. From the studies of three-component diagram for glass formation in these multicomponent systems, one may be able to get some information on the role of cations, especially of such intermediate ions as  $\text{Ti}^{4+}$ , in the glass structure. It may also be worthwhile to determine the most favorable conditions for the glass formation in multicomponent phosphate systems.

In this work, the regions for glass formation in the three-component diagrams were determined for three kinds of phosphate systems containing  $\text{TiO}_2$ —the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$ ,  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$ , and  $\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$  systems. Based on the results of these ternary phosphate systems containing  $\text{TiO}_2$  and that of the  $\text{Na}_2\text{O}-\text{BaO}-\text{P}_2\text{O}_5$  system, the conditions for the formation of the glasses containing maximum amounts of  $\text{TiO}_2$  and  $\text{BaO}$  were investigated for the four-component system of  $\text{Na}_2\text{O}-\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$ .

### Experimental

**Preparation of Glass.** The raw materials used were sodium dihydrogenphosphate dihydrate, sodium triphosphate, anhydrous sodium carbonate, potassium dihydrogenphosphate, anhydrous potassium carbonate, barium carbonate, titanium dioxide, and orthophosphoric acid (85%). All the chemicals except sodium triphosphate were of a reagent grade. Commercial reagent-grade sodium triphosphate was purified before use by recrystallization and dehydration. Mixtures of from two to four kinds of the raw materials were placed in platinum evaporating dishes. For the batch of  $\text{Na}_2\text{O}/\text{P}_2\text{O}_5$  or  $\text{K}_2\text{O}/\text{P}_2\text{O}_5=1$  in the  $\text{Na}_2\text{O}-\text{P}_2\text{O}_5$  or  $\text{K}_2\text{O}-\text{P}_2\text{O}_5$  two-component system,  $\text{NaH}_2\text{PO}_4 \cdot 2\text{H}_2\text{O}$  or  $\text{KH}_2\text{PO}_4$  was used respectively. All the batches were about 15 g in weight.

Samples were heated in a Siliconit electric furnace BSH-1530 (Siliconit Konetsukogyo Co.) equipped with a platinum-platinum-rhodium thermocouple with an automatic thermoelectric regulator. It took about 5 hr to heat them up to

1350 °C, and this temperature was kept for 1 hr. The melt was quenched by pressing with copper plate cooled with water. In the batches containing  $\text{TiO}_2$  with more than 50 mol%  $\text{P}_2\text{O}_5$ , except for the composition in which the homogeneous glass was formed, the crystalline material was separated out in the melt. The homogeneity of the glass formed was immediately observed by the naked eye and under a microscope or a polarizing microscope. No phase separation of the melt itself was found in any of the systems.

The phosphate glass usually contains a small amount of water in its structure when prepared by dehydration and condensation.<sup>3,4)</sup> In addition, the composition of the glass is varied by the loss of  $\text{P}_2\text{O}_5$  during the heating, it depends on the conditions of preparation.<sup>3)</sup> In the  $\text{Na}_2\text{O}-\text{BaO}-\text{P}_2\text{O}_5$  system, the amount of the volatilization loss of  $\text{P}_2\text{O}_5$  was in the range from 3.3 to 14.0 mol%. Taking these facts into account, all the diagrams for the glass formation were expressed in terms of the batch composition.

**Examination of Chemical Durability.** Glass grains (0.25—1 mm) of about 0.5 g were weighed and each portion was placed in a 100-ml Erlenmeyer flask. Then 25 ml of water, 25 ml of 35%  $\text{HCl}$ , and 25 ml of 10%  $\text{NaOH}$  were added separately into each flask, and they were stoppered. After having been shaken at speed of 80 strokes per minute in the thermostat at 25 °C for 1 hr, the weight loss (%) of each sample was determined.

### Results and Discussion

The experimental results for the glass-forming region in each system are given in Figs. 1, 3, and 5—11. Figures 7—11 show the glass-forming region in the  $\text{Na}_2\text{O}-\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$  four-component system. In the systems containing  $\text{TiO}_2$ , the glasses were colored purple or brown due to the titanium ion reduced. The chemical durabilities of the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  and  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  glasses are shown in Figs. 2 and 4.

**$\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  System (Fig. 1).** A slight amount of glass was obtained in the system with more than 50 mol%  $\text{P}_2\text{O}_5$ . However, glass with a large amount of  $\text{TiO}_2$  was obtained by decreasing the amount of  $\text{P}_2\text{O}_5$  to less than 50 mol%. It may be seen in Fig. 1 that the system containing  $\text{Na}_2\text{O}$  and  $\text{P}_2\text{O}_5$  in the ratio of unity is most favorable for formation of the glass with a high  $\text{TiO}_2$  content. This fact may possibly be interpreted by associating the increase in the glass-forming ability of the  $\text{Ti}^{4+}$  ion with the ring-opening polymerization of sodium trimetaphosphate ( $\text{Na}_3\text{P}_3\text{O}_9$ ).<sup>5)</sup> The composition of the glass containing a maximal

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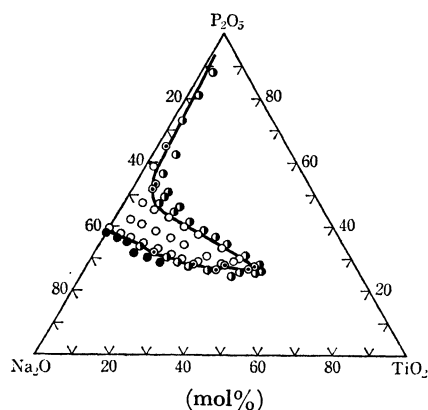


Fig. 1. Glass-forming region in  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  system

○: homogeneous glass, ⊙: trace amounts of crystal in glass, ◐: glass and crystal, ●: crystal.

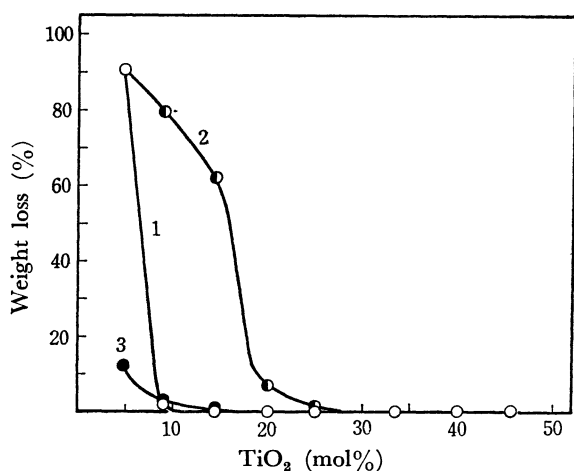


Fig. 2. Chemical durability of  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  glass ( $\text{Na}_2\text{O}/\text{P}_2\text{O}_5=1$ )

1: water, 2: 35% HCl, 3: 10% NaOH.

quantity of  $\text{TiO}_2$  is  $\text{Na}_2\text{O}$ : 27.3,  $\text{TiO}_2$ : 45.5, and  $\text{P}_2\text{O}_5$ : 27.3 mol%.

Figure 2 shows that the chemical durability of the glass is very much increased by increasing the amount of  $\text{TiO}_2$ . The glass containing more than 20 mol%  $\text{TiO}_2$  is insoluble or only slightly soluble in either water, 35% HCl, or 10% NaOH.

**$\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  System (Fig. 3).** With respect to more than 50 mol%  $\text{P}_2\text{O}_5$ , the region of glass formation of this system is similar to that of the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  system. As is shown with arrow marks in Fig. 3, the region of glass formation in the composition along the line of  $\text{K}_2\text{O}/\text{P}_2\text{O}_5=1$  is narrow. This is a marked difference from the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  system, it may be interpreted in terms of the formation of straight-chain molecules of potassium metaphosphate.<sup>6)</sup>

Rao<sup>7,8)</sup> reported a formation of glass in the  $\text{K}_2\text{O}-\text{TiO}_2$  system. Under the experimental conditions in the present work, however, no homogeneous glass was obtained in the same system.

In the A and B regions in Fig. 3, the maximum concentrations of  $\text{TiO}_2$  were about 38 mol% and 35 mol% respectively. The coloration in the glass was observed in only the A region. As is shown in Fig. 4,

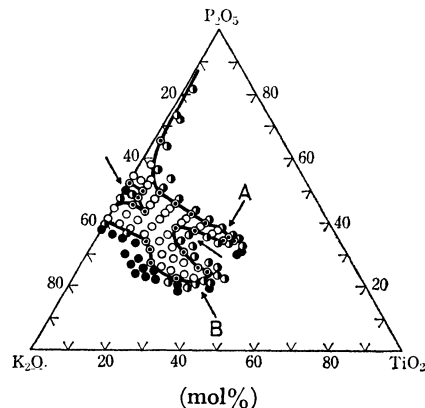


Fig. 3. Glass-forming region in  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  system.

A; Region obtained the glass containing  $\text{TiO}_2$  in the largest quantity in  $\text{K}_2\text{O}/\text{P}_2\text{O}_5 < 1$   
B: Region obtained the glass containing  $\text{TiO}_2$  in the largest quantity in  $\text{K}_2\text{O}/\text{P}_2\text{O}_5 > 1$

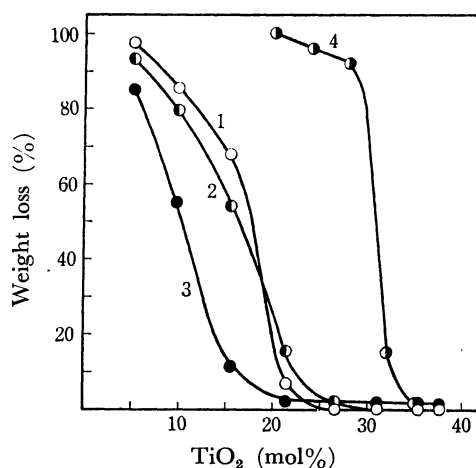


Fig. 4. Chemical durability of  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  glass.

1:  $\text{K}_2\text{O}/\text{P}_2\text{O}_5=0.83$ , water, 2:  $\text{K}_2\text{O}/\text{P}_2\text{O}_5=0.83$ , 35% HCl, 3:  $\text{K}_2\text{O}/\text{P}_2\text{O}_5=0.83$ , 10% NaOH, 4:  $\text{K}_2\text{O}/\text{P}_2\text{O}_5=1.67$ , water.

the increase in the chemical durability of the glass formed in the system of  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  was also observed with an increase in the  $\text{TiO}_2$  content, as in the case of the  $\text{Na}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  system

According to Rao,<sup>7)</sup> the  $\text{Ti}^{4+}$  ion is able to act as a network former in silicate glass. The  $-\text{P}-\text{O}-\text{Ti}-$  linkage is probably formed in the phosphate glass. It seems that the increase in the chemical durability of the glass containing a large amount of  $\text{TiO}_2$  results from the formation of such a linkage.

**$\text{BaO}-\text{TiO}_2-\text{P}_2\text{O}_5$  System (Fig. 5).** The homogeneous glass was obtained in the  $\text{BaO}-\text{P}_2\text{O}_5$  system containing BaO up to approximately 58 mol%, but only about 10 mol%  $\text{TiO}_2$  could be introduced in this glass at a maximum. This is very different from the glass formation in the  $\text{NaO}-\text{TiO}_2-\text{P}_2\text{O}_5$  and  $\text{K}_2\text{O}-\text{TiO}_2-\text{P}_2\text{O}_5$  systems (Figs. 1 and 3).

According to Kordes *et al.*,<sup>9)</sup> the  $\text{BaO}-\text{P}_2\text{O}_5$  glass is classified as a "normal glass" among the binary glasses of alkaline earth oxides and  $\text{P}_2\text{O}_5$ , and the  $\text{Ba}^{2+}$  ion has been regarded as a network modifier. However, the

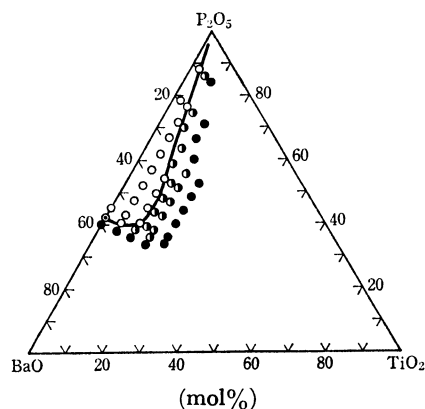


Fig. 5. Glass-forming region in BaO-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> system.

electrostatic bond strength of the divalent Ba<sup>2+</sup> ion, *i.e.*, the ratio of the charge number of the ion to its coordination number, is larger than that of the univalent Na<sup>+</sup> or K<sup>+</sup> ion. Thus it seems that BaO stabilizes the glass structure much more than Na<sub>2</sub>O or K<sub>2</sub>O in the phosphate glasses. The maximum content of TiO<sub>2</sub> in the Na<sub>2</sub>O-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> glass was found to be 45.5 mol% at Na<sub>2</sub>O/P<sub>2</sub>O<sub>5</sub>=1, but that in the BaO-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> glass only about 10 mol% at BaO/P<sub>2</sub>O<sub>5</sub>=1 (Figs. 1 and 5). The Ba<sup>2+</sup> ion probably interferes with the introduction of the tetravalent Ti<sup>4+</sup> ion into the glass structure.

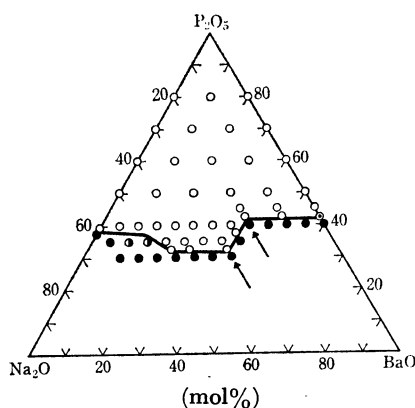


Fig. 6. Glass-forming region in Na<sub>2</sub>O-BaO-P<sub>2</sub>O<sub>5</sub> system.

**Na<sub>2</sub>O-BaO-P<sub>2</sub>O<sub>5</sub> System (Fig. 6).** This system consists of two kinds of network modifiers (Na<sub>2</sub>O and BaO) and a network former (P<sub>2</sub>O<sub>5</sub>).

As is shown in Fig. 6, the glass is obtained over a considerably wide region. The lower limit of the P<sub>2</sub>O<sub>5</sub> content in the glass formation is approximately 32 mol%. From the boundary curve in Fig. 6, it is known that the P<sub>2</sub>O<sub>5</sub> content is constant in the regions of the Na<sub>2</sub>O contents up to 20 mol%, 30–40 mol%, and 50–60 mol%, whereas the BaO content is constant in the region of 20 to 30 mol% Na<sub>2</sub>O.

This shape of the boundary curve may possibly be interpreted as follows: the boundary of glass formation may be determined by the difference in the action between Na<sup>+</sup> and Ba<sup>2+</sup> ions in the glass structure at the glass-forming limit. Thus, the increase in the

Na<sub>2</sub>O contents from 20 up to 30 mol% along the boundary (arrow marks in Fig. 6) seems to be the result of a break in the -P-O-P- linkage rather than of the substitution of the Ba<sup>2+</sup> ion by the Na<sup>+</sup> ion. In the glass-forming boundary, the Ti<sup>4+</sup> ion may be introduced quite easily into this glass when the Na<sub>2</sub>O content is 20–30 mol%.

**Na<sub>2</sub>O-BaO-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> System (Figs. 7–11).** In order to observe the effect of Na<sub>2</sub>O on the glass forma-

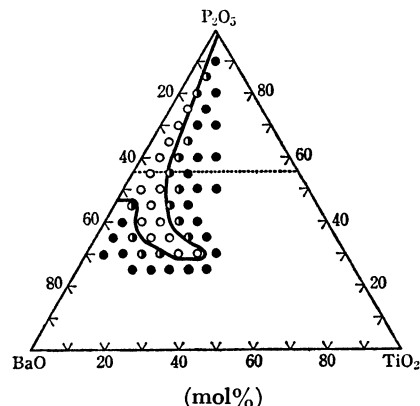


Fig. 7. Glass-forming region in Na<sub>2</sub>O-BaO-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> system containing 10 mol% Na<sub>2</sub>O (dotted line represents P<sub>2</sub>O<sub>5</sub> content is 50 mol%).

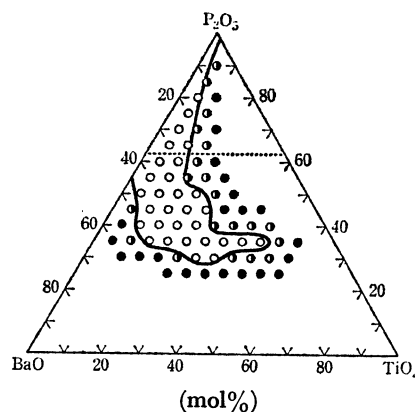


Fig. 8. Glass-forming region in Na<sub>2</sub>O-BaO-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> system containing 20 mol% Na<sub>2</sub>O (dotted line represents P<sub>2</sub>O<sub>5</sub> content is 50 mol%).

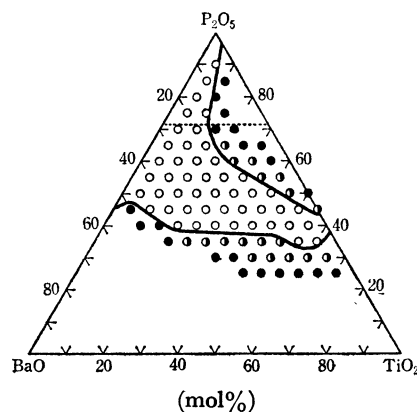


Fig. 9. Glass-forming region in Na<sub>2</sub>O-BaO-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> system containing 30 mol% Na<sub>2</sub>O (dotted line represents P<sub>2</sub>O<sub>5</sub> content is 50 mol%).

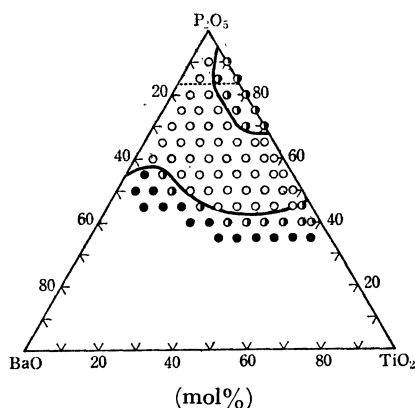


Fig. 10. Glass-forming region in  $\text{Na}_2\text{O}$ - $\text{BaO}$ - $\text{TiO}_2$ - $\text{P}_2\text{O}_5$  system containing 40 mol%  $\text{Na}_2\text{O}$  (dotted line represents  $\text{P}_2\text{O}_5$  content is 50 mol%).

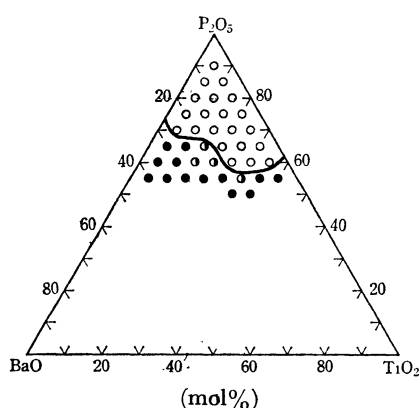


Fig. 11. Glass-forming region in  $\text{Na}_2\text{O}$ - $\text{BaO}$ - $\text{TiO}_2$ - $\text{P}_2\text{O}_5$  system containing 50 mol%  $\text{Na}_2\text{O}$ .

tion in the  $\text{Na}_2\text{O}$ - $\text{BaO}$ - $\text{TiO}_2$ - $\text{P}_2\text{O}_5$  system, the glass-forming region was studied for batches with different  $\text{Na}_2\text{O}$  contents (10, 20, 30, 40 and 50 mol%).

The results are shown in Figs. 7–11. The region in which the glass containing  $\text{TiO}_2$  is formed extends up to 30 mol% of  $\text{Na}_2\text{O}$ . The decrease in the  $\text{P}_2\text{O}_5$  content

causes an increase in the  $\text{TiO}_2$  content and a decrease in the  $\text{BaO}$  content in the glass. In addition, the tendency of the glass-forming region to be extended was observed in the system of up to 30 mol%  $\text{Na}_2\text{O}$  along with the compositions of the  $\text{P}_2\text{O}_5$  contents of 27–28 mol%. The  $\text{Ti}^{4+}$  ion seems to be introduced easily into the glass structure at this amount of  $\text{P}_2\text{O}_5$ . The  $\text{P}_2\text{O}_5$  content corresponds to the lower limit of the  $\text{P}_2\text{O}_5$  contents in the  $\text{Na}_2\text{O}$ - $\text{TiO}_2$ - $\text{P}_2\text{O}_5$  system (about 27 mol%). The glass-forming region in the  $\text{Na}_2\text{O}$ - $\text{BaO}$ - $\text{TiO}_2$ - $\text{P}_2\text{O}_5$  system (Figs. 7–11) extends to the boundary line in the  $\text{Na}_2\text{O}$ - $\text{TiO}_2$ - $\text{P}_2\text{O}_5$  system (Fig. 1) when the  $\text{Na}_2\text{O}$  content is 27–28 mol%.

These results suggest that a certain amount of  $\text{Na}_2\text{O}$  is necessary for the formation of the glass containing a large amount of  $\text{TiO}_2$  in the presence of  $\text{BaO}$ . This amount is probably in the range from 27 to 28 mol%  $\text{Na}_2\text{O}$ . Thus, it may be concluded that the  $\text{Ti}^{4+}$  ion is easily introduced into the glass when both  $\text{Na}_2\text{O}$  and  $\text{P}_2\text{O}_5$  are 27–28 mol%;  $\text{Na}_2\text{O}/\text{P}_2\text{O}_5 \approx 1$ . These results also support the interpretation of the boundary of glass formation in the  $\text{Na}_2\text{O}$ - $\text{BaO}$ - $\text{P}_2\text{O}_5$  system (Fig. 6).

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